

Structural, Morphological, and Optical Studies of ZnS-MoS₂ Nanocomposite via Hydrothermal Route

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ZnS-MoS₂ nanocomposite was synthesized by a cost-effective and eco-friendly hydrothermal method to investigate and compare the structural and optical properties of ZnS-MoS₂ nanocomposite powder with those of pure ZnS and MoS₂. The structural, morphological, and optical absorption properties of the prepared powder samples were characterized by powder X-ray diffraction (XRD), scanning electron microscopy (SEM), diffuse reflectance spectroscopy (DRS), and Fourier transform infrared (FT-IR) spectroscopy. Powder XRD results showed both cubic-phase and hexagonal-phase structures in ZnS and MoS₂. The average crystallite size varied from 20 nm to 25 nm for the hydrothermally prepared nanocomposite. The dislocation density and lattice strain were also calculated. The surface morphology of the prepared nanocomposite showed spherical, hexagonal, and flower-shaped nanoparticles with agglomeration. With the coupling of ZnS and MoS₂, a slight variation was found in the energy band gap that slightly red-shifted from 3.61 to 3.57 eV. The characteristic frequencies related to ZnS-MoS₂ nanocomposite and other related precursor molecules were determined by FT-IR spectra.

Keywords: ZnS, MoS₂, Hydrothermal method, XRD, FT-IR, DRS

INTRODUCTION

Nanotechnology is a cutting-edge technology that emerged in the twenty-first century. Manipulating the structure of materials to transform them into nanoscale structures changes their properties and leads to the development of new principles. Over the past three decades, theoretical and experimental calculations have been used to study the UV-Vis-NIR spectra of semiconductor nanoparticles due to their distinctive size-dependent properties and the quantum mechanical phenomena in semiconductors. Any variation in the size of particles can cause considerable changes in their electronic, structural, and optical properties. This is due to the three-dimensional (3-D) quantum confinement of holes and electrons that

occurs when the size of the particle approaches the exciton Bohr radius, which is the separation distance between the electron and hole. Semiconductors with a tunable energy band gap are widely considered to be the materials for next-generation flat-panel displays, optoelectronic devices, sensors, photovoltaic devices, photonic band-gap devices, laser, etc.

Among various semiconductors, molybdenum disulfide (MoS₂), a two-dimensional transition metal dichalcogenide semiconductor with a direct band gap, has a distinctive layered structure and some unique properties, such as high electrical conductivity, high electron mobility, high chemical stability, and large surface area [1,2]. MoS₂ has several applications in the field of optoelectronics due to its excellent light absorption to the visible and IR regions. This absorption property of MoS₂ can be attributed to its band gap, which varies from around 1.3 eV to 1.8 eV [3]. In

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addition, the layer-dependent optoelectronic and electronic properties of MoS₂ allow the tailoring of its properties as per the required application. Notwithstanding that, MoS₂ has some shortcomings, such as quick recombination of photogenerated hole-electron pairs and slow electron transfer rate. To overcome these drawbacks, MoS₂ has been mixed with other materials having semiconducting nature to enhance its properties and increase its domain of application [4]. As regards ZnS nanophosphor, it also has some special features, such as a wide direct band gap, favorable transport properties, and high electron mobility [5]. In addition to these interesting properties, ZnS nanophosphor has adequate thermal stability and high transmittance in the infrared range, making it a promising material in optoelectronics [6]. Both ZnS and MoS₂ can be used as sulfur sources with similar lattice parameters [7]. It has been reported that ZnS-MoS₂ nanocomposite exhibited improved structural and optical properties compared with either of them individually [4]. The ZnS-MoS₂ nanocomposite has applications in both photocatalysis, due to the band gap difference in its counterparts (*i.e.*, ZnS and MoS₂), and optoelectronics, due to its wide band gap. The objective of the present study was to investigate the structural and optical properties of ZnS-MoS₂ nanocomposite powder and compare them with those of pure ZnS and pure MoS₂ nanopowders. These two compounds, namely, ZnS and MoS₂, were selected because both had wide applications in photocatalysis and optoelectronic devices. A composite containing 60% ZnS and 40% MoS₂ was used in this study since this combination was reported to have wide applications [8]. In the present work, sulfur-based ZnS-MoS₂ nanocomposite was synthesized by the hydrothermal method [9], and its structural and optical properties were characterized using different characterization techniques, including XRD, SEM, DRS, and FT-IR spectral studies.

STARTING MATERIALS AND SYNTHESIS METHOD

Materials

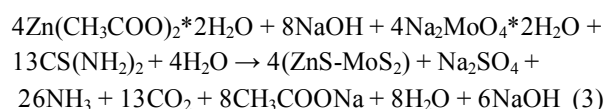
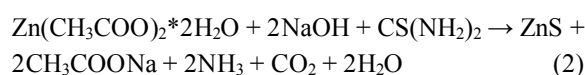
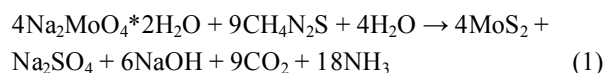
To prepare ZnS-MoS₂ nanocomposite powders, all chemicals with 99.9% purity were used. Sodium molybdate dihydrate (Na₂MoO₄·2H₂O), zinc acetate dihydrate (C₄H₁₀O₆Zn), thiourea (CH₄N₂S), and sodium hydroxide (NaOH) were used as starting materials. Double deionized

water was used in the dilution process and sample synthesis. All the glassware used in the preparation process was cleaned with acetone.

Synthesis of ZnS-MoS₂ Nanocomposite

In this process, ZnS-MoS₂ was prepared with a 3:2 molar ratio using a single-step hydrothermal method. For this purpose, 10 mmol of CH₄N₂S, 1 mmol of Na₂MoO₄·2H₂O, and 1.5 mmol of Zn (CH₃COO)₂·2H₂O were weighed, dissolved into 100 ml of double deionized water in a 250 ml conical flask, and stirred for 2 h using a magnetic stirrer, followed by the addition of 10 ml of 0.2 M NaOH solution drop by drop. NaOH solution was added to maintain the chemical stability at alkaline pH. After stirring, the stirred chemical was transferred into a Teflon-lining autoclave with a capacity of 150 ml, heated under autogenous pressure, placed in a hot air oven at 180 °C for 12 h, and allowed to return to room temperature for 12 h, which led to the formation of a black color precipitate. This precipitate was centrifuged at 10,000 rpm and washed four times with ethyl alcohol and double deionized water alternatively. The final precipitate was dehydrated at 120 °C for 6 h and ground with an agate mortar to obtain the powder form of the prepared ZnS-MoS₂ nanocomposite. The same procedure was followed to prepare ZnS and MoS₂ using their precursors by maintaining a 1:4 molar ratio of thiourea.

The balanced chemical equations for the synthesis processes of (1) MoS₂, (2) ZnS, and (3) ZnS-MoS₂ nanocomposites are presented below:



CHARACTERIZATION TECHNIQUE

The nature of the prepared ZnS-MoS₂, ZnS, and MoS₂

nanopowder samples was tested at room temperature using a Rigaku MiniFlex 600 diffractometer with Cu K α radiation (wavelength = 1.5406 Å) scanned from 3° to 90°. The dynamic reflection spectra (DRS) of the prepared samples were recorded using a UV/Vis/NIR-spectrophotometer (SPECORD-210 plus, Analytik Jena) in the region 200-1000 nm. The morphology and elemental analysis mapping of the prepared nanopowders were carried out using a JEOL JSM-IT500 scanning electron microscope (SEM) with energy dispersive X-ray analysis (EDAX). The vibrational IR spectra, in wavenumber region from 4000 to 400 cm⁻¹ unit, of the prepared samples were recorded on Bruker ALPHA II model FT-IR equipment using the KBr pellet technique.

RESULTS AND DISCUSSION

Powder XRD Studies

XRD pattern was used to obtain some information about the phase purity of the prepared ZnS-MoS₂ nanocomposite powder. The recorded diffraction patterns in the range of 3° to 90° are shown in Fig. 1. The XRD pattern of ZnS-MoS₂ nanocomposite showed the presence of both cubic and hexagonal structures in ZnS and MoS₂, and the diffraction pattern peaks at 28.66°, 33.18°, 47.62°, 56.49°, 59.24°, 69.58°, 76.87°, 79.15°, and 88.62° corresponded to the lattice planes (111), (200), (220), (311), (222), (400), (331), (420), and (422) of zinc blende cubic structure of ZnS, as reported in the JCPDS card (No: 01-077-2100). In addition to the above peaks, the XRD pattern related to the ZnS-MoS₂ nanocomposite exhibited peaks at 2 θ values of 26.85°, 27.26°, 28.18°, 29.21°, and 33.56°, which corresponded to the (100), (102), (103), (104), and (107) diffraction peaks of Wurtzite-10H phase of ZnS, as reported in the JCPDS card (Number: 00-012-0688). Moreover, a prominent peak at 2 θ = 14.02° corresponding to the (002) of the hexagonal phase of MoS₂ matched the XRD data of the standard ICDD-JCPDS card (Number: 01-075-1539) [10-12].

Undoped ZnS showed a cubic phase, but when it was coupled with MoS₂, it exhibited both cubic and hexagonal phases. Due to the coupling between ZnS and MoS₂, the cubic phase peaks intensity of ZnS decreased and the hexagonal phase peaks of ZnS increased. It can be

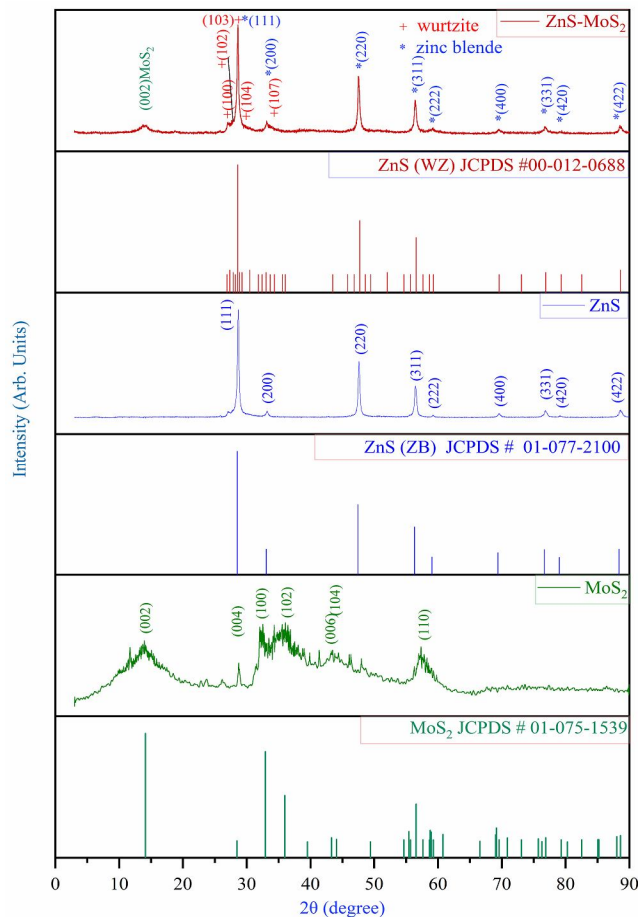


Fig. 1. The XRD pattern of ZnS, MoS₂, and ZnS-MoS₂ nanoparticles.

understood that the coupling effect heavily influenced the local distortions in the host lattice structure of the ZnS-MoS₂ sample [13]. The XRD pattern also revealed intense peaks with broad nature, which, in turn, indicates that the crystallite sizes of the synthesized powders were in the order of nanometer range. The slight shift in peak position represents strong evidence for the existence of the local disorder and lattice strain in the nanocomposite.

The mean size of the crystallite (D) can be measured from the FWHM-full width value at half maximum of intensity related to diffraction peaks. The Debye-Scherrer's equation was adapted to estimate the average crystallite size (D) of the hydrothermally prepared samples.

The widely used Debye-Scherrer's crystallite size

equation is presented below:

$$D = \frac{S_f \lambda}{(\cos \theta) \beta}$$

Here, S_f ($= 0.9$) is shape factor, D denotes the crystallite size, λ is the length of the wave (1.5406 \AA) of the incident (Cu-K α) X-Ray beam, β and θ (both are in radians) are the FWHM-full width value at half maximum of intensity and angle of diffraction, respectively [14].

Also, the δ -dislocation density and ϵ -microstrain were calculated using Williamson-Smallman relation $\delta = \frac{1}{D^2}$ and Stokes-Wilson relation $\epsilon = \frac{\beta}{4 \tan \theta}$, respectively [15]. The calculated values of crystallite size, micro-strain, and dislocation densities for the prepared samples ZnS, MoS₂, and ZnS-MoS₂ are shown in Table 1.

SEM and EDAX Studies

SEM studies were carried out to investigate surface morphologies and nanoparticle distribution of the prepared nanopowder samples under normal ambient temperature and pressure. Figure 2 shows SEM micrographs of ZnS, MoS₂, and ZnS-MoS₂ nanocomposites, respectively. While the SEM images of ZnS show hexagonal spheres, those of MoS₂ show spherical flowers type morphology. The coupling of ZnS and MoS₂ (*i.e.*, ZnS-MoS₂ nanocomposite) shows spherical hexagonal flowers with agglomerations.

The composition of MoS₂ nanoparticles, ZnS nanoparticles, and ZnS-MoS₂ nanocomposite was confirmed by the EDAX technique, which revealed the existence of the desired elements (*i.e.*, S, Mo, and Zn) in the prepared samples. Figure 3 depicts the examined EDAX patterns of the respective prepared samples.

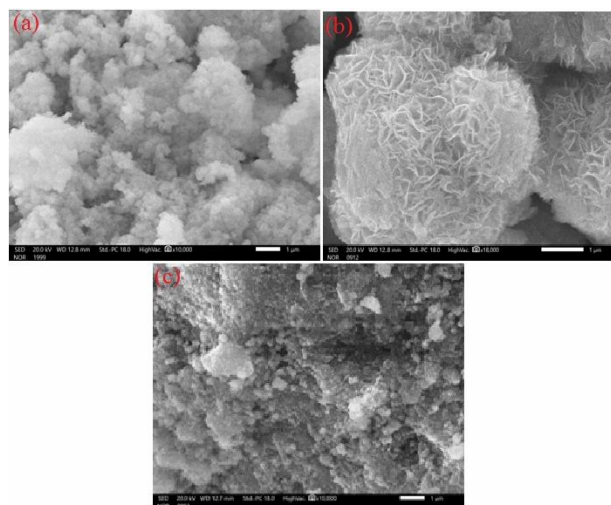


Fig. 2. The morphological SEM images of ZnS (a), MoS₂ (b), and ZnS-MoS₂ (c).

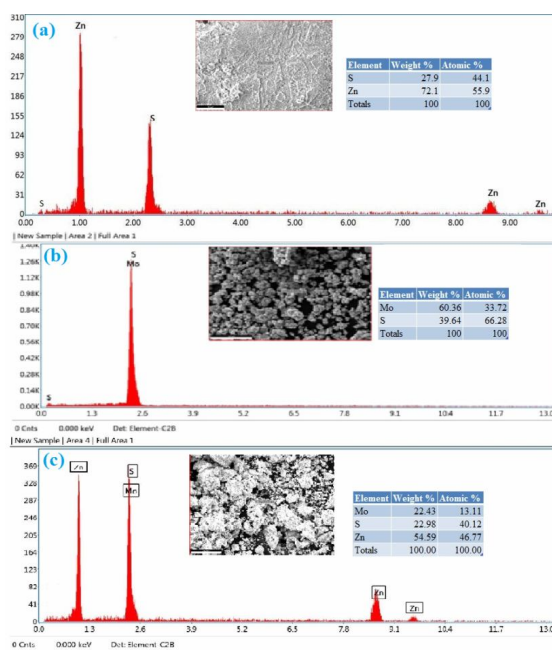


Fig. 3. The EDAX patterns of ZnS (a), MoS₂ (b), and ZnS-MoS₂ (c).

Table 1. Mean Crystallite Size, Micro-Strain and Dislocation Densities of MoS₂, ZnS, and ZnS-MoS₂ Nanoparticles

Sample	Position (2 θ -degree)	FW-HM (β -degree)	Crystallite size (D) (nm)	Lattice strain $[(\epsilon) \times 10^{-4}]$	Dislocation density (δ) (lines/m ²)
MoS ₂	14.02	0.4598	17.41	163.36	3.29×10^{15}
ZnS	28.66	0.3529	23.25	60.28	1.85×10^{15}
ZnS-MoS ₂	28.60	0.4170	19.67	71.39	2.58×10^{15}

Optical Properties

DRS is a very good mechanism for samples in the powdered form as it does not require any suitable medium for measurement. The band gap calculations were performed using the observed reflection spectra from 200 to 1000 nm wavelength region. The band gap of the synthesized nanoparticles ZnS, MoS₂, and ZnS-MoS₂ were estimated using the Kubelka-Munk equation:

$$F(R) = \frac{(1-R)^2}{2R}$$

Here, R is the reflectivity of the sample.

Linear absorption coefficient α can be obtained from F(R) value by a transformation [16], as shown below:

$$\alpha = \frac{F(R)}{t} = \frac{\text{Absorbance}}{t}$$

where t is proportionality constant. Absorption co-efficient (α) and band gap (E_g) are connected by a relation ($\alpha h\nu$) = A($h\nu - E_g$)ⁿ; ($\alpha h\nu$) = F(R) hν; A($h\nu - E_g$)ⁿ = F(R) hν and

[F(R)hν]^{1/n} = A($h\nu - E_g$); here the values of n = 1/2, 3/2, 2, and 3 show allowed direct, direct forbidden, allowed indirect, and indirect forbidden transitions [17].

Figure 4 represents a graph between (F(R)hν)² and hν. The band gap values were extrapolated from the graph for the samples ZnS, MoS₂ (inset plot), and ZnS-MoS₂. This is an excellent method to characterize the semiconducting nanoparticles which show quantum size phenomenon due to pairs of electrons and holes generated by light [18]. The reflection spectrum in the UV-Vis region depends on the actual size of the semiconductor nanoparticles. The maxima of reflection decrease with nanoparticles size. MoS₂ with a 1.34 eV energy band gap had a NIR reflection band, and ZnS with a 3.61 eV energy band gap had an intensive UV reflection band. ZnS-MoS₂ also responded to the UV region with a 3.57 eV energy band gap, though with a redshift. The band gap values extrapolated from the graph are shown in Table 2.

FT-IR Study

To confirm the existence of precursor residuals and

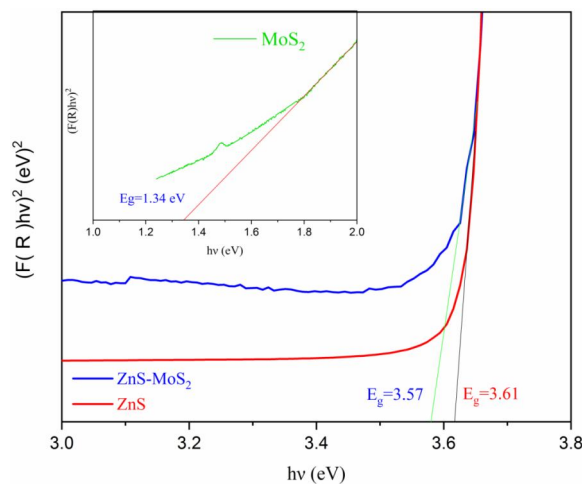


Fig. 4. The Munk plot for ZnS, MoS₂ (inset graph) and ZnS-MoS₂ nanoparticles.

Table 2. The Band Gap Values of ZnS, MoS₂, and ZnS-MoS₂ Nanoparticles

Sample code	Band gap (eV)
ZnS	3.61
MoS ₂	1.34
ZnS-MoS ₂	3.57

inspect the presence of various characteristic functional groups in MoS₂, ZnS, and ZnS-MoS₂ nanopowder samples, FT-IR measurements were performed in 400-4000 cm⁻¹ range in the mode of transmittance, and their FT-IR spectra are illustrated in Fig. 5. The noticed peaks were thoroughly analyzed, and the peak positions, along with their characteristic nature, are summarized in Table 3.

At 3420 cm⁻¹, a broad dip occurred due to the O-H stretching caused by the strong interaction between H₂O and ZnS lattice surface. This happens due to the residuals of precursors during the characterization process or during the synthesis [19]. Around 2064-2080 cm⁻¹, peaks are due to -CH₂ stretching [20] and at around 1620-1637 cm⁻¹ due to H-O-H bending mode [19]. The presence of H₂O was confirmed by its bending vibration at around 1620-1637 cm⁻¹.

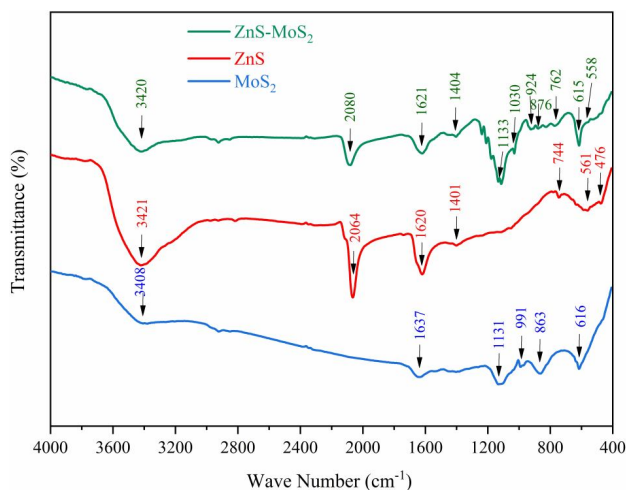


Fig. 5. The FT-IR spectra for pure ZnS, pure MoS₂, ZnS-MoS₂ nanocomposites.

Bands observed near 1404-1401 cm⁻¹ can be attributed to the C-H stretching [21]. The bands located near 1133-991 cm⁻¹ confirm the complex sulfur formation in MoS₂ with active sites [22]. The bands around 924, 762, 744, 558, and 476 cm⁻¹ can be attributed to Zn-S vibration [23]. The bands near 876, 863, 616, and 615 cm⁻¹ can be attributed to

S-Mo-S vibration [24].

CONCLUSIONS

ZnS, MoS₂, ZnS-MoS₂ nanocomposites were prepared via a hydrothermal route. The XRD results showed that pure ZnS had both cubic (ZB) and hexagonal (WZ) phases, but when it was coupled with MoS₂, the hexagonal phase increased. The mean crystallite size of the hydrothermally synthesized nanocomposite was around 20-25 nm. The morphology of the prepared nanocomposite showed spherical hexagonal flowers with agglomerations. The coupling of ZnS-MoS₂ caused a slight energy band gap variation that red-shifted from 3.61 to 3.57 eV. FT-IR spectra revealed the possible stretching and bending vibrations modes of the prepared samples.

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Table 3. Peak Position Assignment and the Corresponding Wave Numbers in IR Spectra of ZnS, MoS₂, and ZnS-MoS₂ Nanocomposites

Vibrational frequency (cm ⁻¹)			Assignment of various bands
ZnS	MoS ₂	ZnS-MoS ₂	
476	-	-	Zn-S vibration
561	-	558	Stretching mode vibrations of Zn-S bonds
-	615	616	S-Mo-S bonds vibrations (stretching mode)
744	-	762	Stretching mode vibrations of Zn-S bonds.
-	863	876	S-Mo-S bonds vibrations (stretching mode)
-	-	924	Zn-S vibrations
-	991	1030	MoS ₂ active sites
-	1131	1133	MoS ₂ active sites
1401	-	1404	C-H stretching
1620	1637	1621	H ₂ O molecules on the particle surface
2064	-	2080	CH ₂ stretching vibration
3421	3408	3420	Stretching vibration of bonded and non-bonded -O-H groups

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CONFLICTS OF INTEREST

The authors declare no conflict of interest.

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